



# Electrodeposition of Cu<sub>2</sub>O Nanowires Using Nanoporous Alumina Template

Jaeho Oh,<sup>a</sup> Yongsug Tak,<sup>a,\*</sup> and Jaeyoung Lee<sup>b,z</sup>

<sup>a</sup>Department of Chemical Engineering, Inha University, Incheon 402-751, South Korea

<sup>b</sup>Research Institute of Industrial Science and Technology, Pohang 790-330, South Korea

We first electrodeposited highly ordered Cu<sub>2</sub>O nanowires using a home-made alumina template. Home-made nanoporous alumina templates were prepared by a two-step anodization method and selective removal of aluminum substrate by bromine solution. Chronopotentiometry (*i.e.*, constant current method) of  $-0.5 \text{ mA/cm}^2$  was applied to deposit compact Cu<sub>2</sub>O nanowires without disconnection into the honeycomb shaped pores of a home-made nanoporous alumina membrane. Analyzing morphological observations by scanning electron microscopy and transmission electron microscopy, we observed that the diameter of Cu<sub>2</sub>O nanowire was *ca.* 57 nm and the length was 4  $\mu\text{m}$  demonstrating an aspect ratio of 70. X-ray diffraction structural analysis indicated that Cu<sub>2</sub>O(200) was preferentially formed compared with other Cu<sub>2</sub>O crystalline phases which may be due to an adjusted solution pH 9.

© 2004 The Electrochemical Society. [DOI: 10.1117/1.1640637] All rights reserved.

Manuscript submitted July 11, 2003; revised manuscript received August 19, 2003. Available electronically January 22, 2004.

## Experimental

To create hexagonally ordered nanoporous alumina membrane, we used a two-step anodization method followed by selective dissolution of aluminum and barrier aluminum oxide (Fig. 1). A high purity aluminum sheet (Aldrich, 99.999% purity, 0.25 mm thick) was used in this experiment. The specimen was embedded in a Teflon holder with an exposed surface area of 1.18 cm<sup>2</sup> and electrolyte was rigorously mixed and maintained at 0°C. Pretreatment of the aluminum sheet, *i.e.*, electropolishing process was performed by constant-voltage condition of +21 V below 0°C for 4 min in a mixed solution of HClO<sub>4</sub> (Aldrich), C<sub>2</sub>H<sub>5</sub>OH (Aldrich), and deionized water (*v/v* = 8:90:2). The electropolished aluminum foil was then rinsed with isopropyl alcohol and deionized water.

As shown in Fig. 1a, first anodization in 0.3 M oxalic acid (Aldrich) by applying +40 V at 0°C and alumina dissolution by dipping in chromic acid solution were performed in sequence, resulting in a prepatterned aluminum substrate (Fig. 1b). Then, the same condition of the first anodization was applied for a second anodization resulting in an ordered nanopore array (Fig. 1c). To make an alumina membrane defect-free and in the absence of an obstacle in the nanohole of the membrane, complete removal of the barrier layer and the bottom aluminum is needed. Therefore, we used the following three

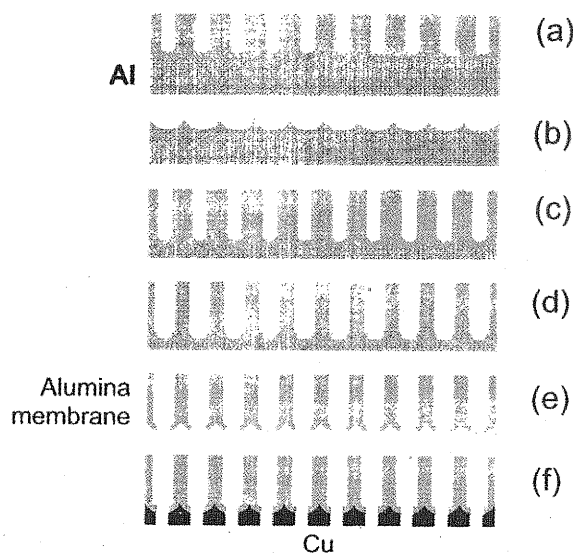


Figure 1. Schematic diagram of the fabrication of the home-made alumina membrane with well-ordered nanosize hole.

Materials with at least one-dimension in the range of 1-100 nm have attracted steadily growing interest owing to their distinct, attractive properties, as well as their unique applications compared with bulk materials.<sup>1-4</sup> Nanowires, *inter alia*, should play an important role as both interconnectors and active components in fabricating nanoscale devices.

Of late, the formation of nanoporous alumina template via the electro-oxidation of aluminum is of great interest in relation to fabrication of ordered nanosized materials in biological, electronic, thermoelectric, and photonic devices.<sup>5-7</sup> This increasing attraction of porous alumina as a template is due mainly to an easy and low-cost manufacturing process. In addition, pore diameters between 4 and 200 nm and interpore distance between 50 and 500 nm of an anodic alumina template could be controlled by applied voltage, solution temperature, and electrolyte concentration.<sup>8-13</sup>

Until very recently, nanowires of metals,<sup>14-16</sup> semiconductors,<sup>17-19</sup> and polymers<sup>20</sup> have been made by several research groups into hexagonally ordered nanoporous alumina template. Particularly, the fabrication of nanotube of polystyrene (PS) with a cylindrical monodisperse diameter by wetting phenomena could be references in a broad range of nanotechnology. Ni nanowires made by a potential modulation mode,<sup>21</sup> known as an attractive tool for the modification and preparation of metal oxide films, have potential applications in high-density magnetic memories. Antimony (Sb)<sup>16</sup> nanowires may be used for the production of metal-semiconductor heterogeneous junction nanowire arrays. Sander *et al.* presented that thermoelectric materials, Bi<sub>2</sub>Te<sub>3</sub> nanowires<sup>17,18</sup> array with controlled aspect ratios were deposited into a nanoporous alumina template. Nanoarrays of TiO<sub>2</sub><sup>19</sup> into commercial Al<sub>2</sub>O<sub>3</sub> membrane (Whatman, Anodisc) with a small pore size of 200 nm were prepared by a simple dipping technique and they showed higher dielectric constant compared with conventional Al<sub>2</sub>O<sub>3</sub> film formed in an aluminum anodizing process.

In this work, we first tried to fabricate cuprous oxide (Cu<sub>2</sub>O) nanowires using a home-made nanoporous alumina template by electrodeposition technique. Transmission electron microscopy (TEM), scanning electron microscopy (SEM), X-ray diffraction (XRD), and Energy-dispersive spectrometry (EDS) analysis clearly showed the formation of Cu<sub>2</sub>O nanowires with a diameter of 57 nm. Cu<sub>2</sub>O is a nonstoichiometric p-type semiconductor with a bandgap of 2.0 eV and recently was proposed as a photocatalytic material for splitting water into H<sub>2</sub> and O<sub>2</sub> via visible light irradiation.<sup>22-24</sup> It is also a potential material for fabrication of low-cost solar cells.<sup>25,26</sup>

\* Electrochemical Society Active Member.  
E-mail: stmlee@yahoo.com

1 at 5 A  
osition:  
00 + 2

rate by  
urchin-  
meters,  
ms.

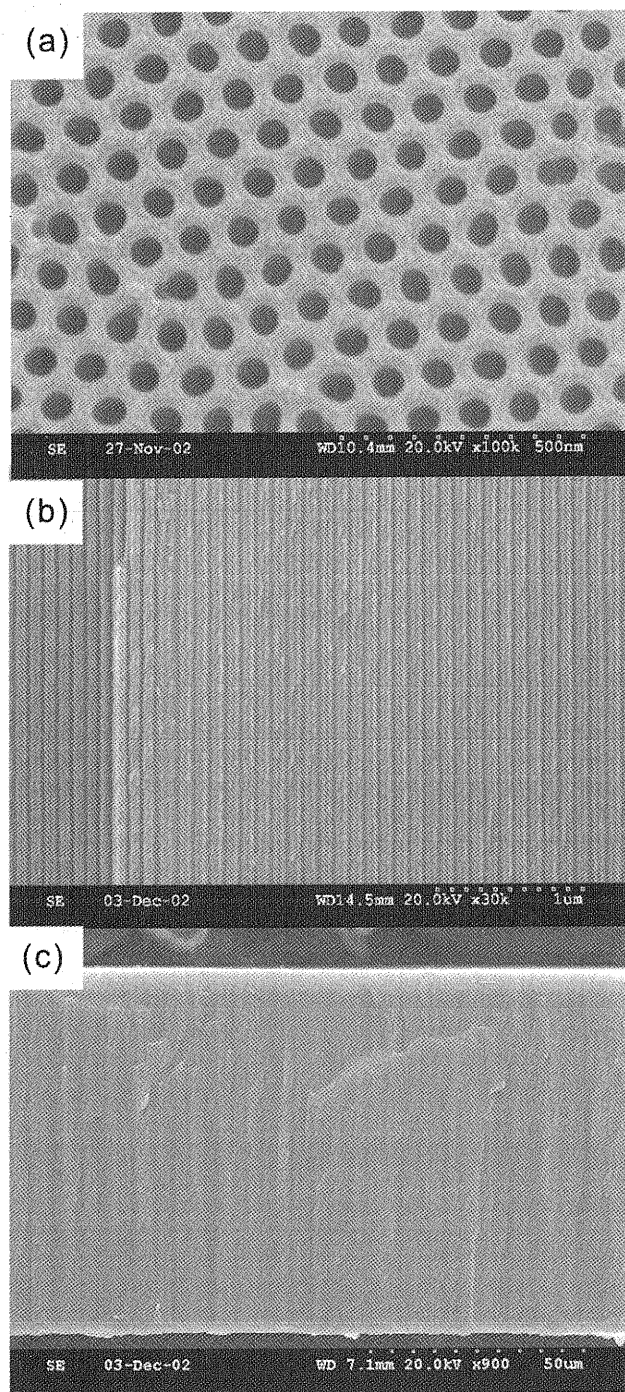
ily dis-  
perature  
dispers-  
chang-  
ompos-  
ed.

istry of

this ar-

36).

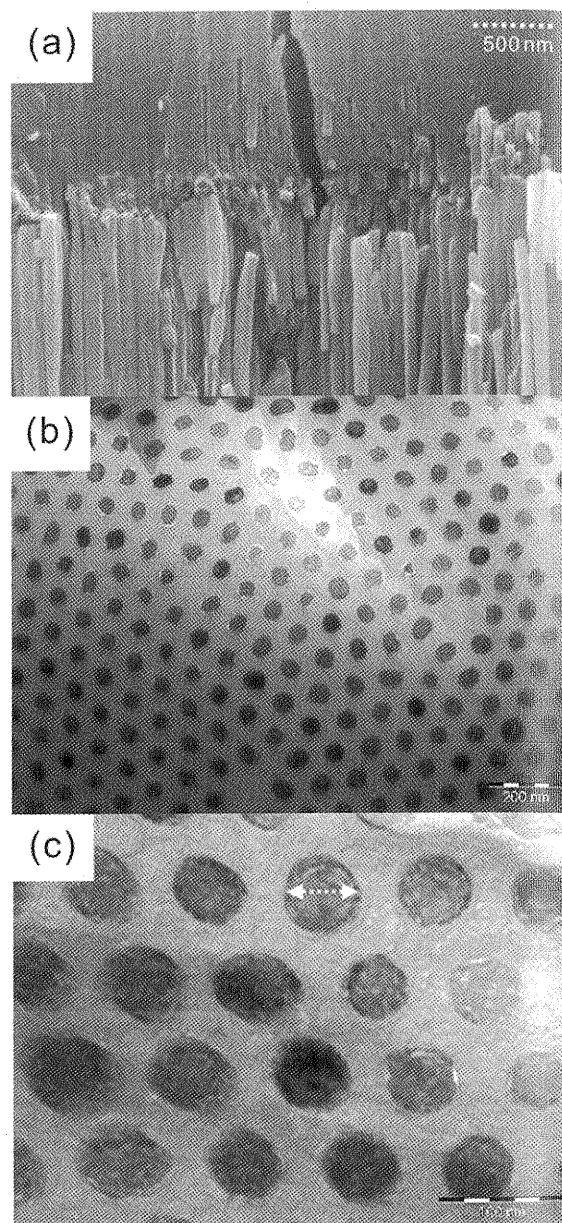
48, 1119



**Figure 2.** SEM image of the home-made alumina membrane with pore distance of 103 nm and pore diameter of 57 nm; (a) top-view and (b) and (c) cross-sectional view in the middle.

processes (Fig. 1d and e); (i) applying 40% of the anodizing voltage (*i.e.*, 16 V) for 20 min, (ii) dipping process in 5 wt %  $\text{H}_3\text{PO}_4$  (Aldrich) solution at 30°C for 30 min, and (iii) removal of the bottom aluminum substrate with bromine solution ( $\text{Br}_2:\text{CH}_3\text{OH}$ (Aldrich) = 1:9, v/v). The dc sputtering technique was applied to form a conducting Cu substrate on one side of the membrane (dark area in Fig. 1f).

$\text{Cu}_2\text{O}$  nanowires were deposited galvanically into a hexagonal nanoarray alumina membrane using the three-electrode system. Counter electrode and reference electrode were Pt mesh and saturated calomel electrode (SCE), respectively. A total of 45 g of



**Figure 3.** Morphological observations of  $\text{Cu}_2\text{O}$  nanowires array composite. (a) SEM data of the cross-sectional view, (b) and (c) TEM top view.

99.9995%  $\text{CuSO}_4$  (Aldrich) was dissolved in 75 mL of 88% lactic acid (Aldrich) and the copper ions stabilized by complexing with lactate ion. 5 M NaOH (Aldrich) aqueous solution was added to adjust the pH to 9. Potentiostat and galvanostat (PAR 273A) was used to apply a current density of  $-0.5 \text{ mA/cm}^2$  and the solution temperature of 65°C was constantly maintained during the experiments.

The surface morphology of  $\text{Cu}_2\text{O}$  nanowires was analyzed by field-emission scanning electron microscopy (FESEM, Hitachi S-4300) and TEM (Philips CM200). The crystallinity and quantitative analysis of prepared  $\text{Cu}_2\text{O}$  nanowires was carried out using XRD (Philips DY616) and EDS coupled with the TEM equipment.

## Results and Discussion

Figure 2a and b shows the top view and cross-sectional view of the home-made alumina membrane with highly ordered nanopores.

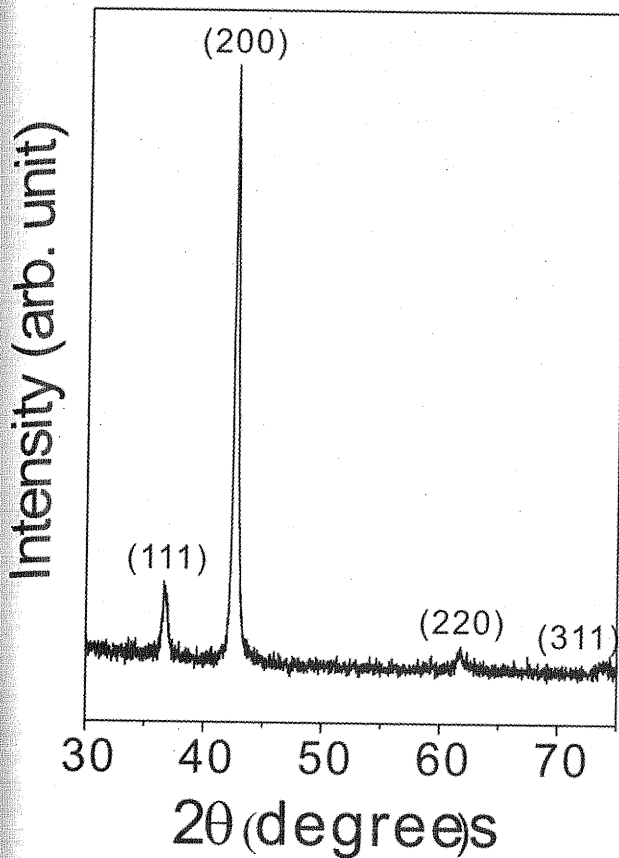


Figure 4. XRD pattern of a sample of  $\text{Cu}_2\text{O}$  nanowires in the alumina template prepared by the electrodeposition method at  $65^\circ\text{C}$ .

The diameter, an interpore distance, and the thickness of aluminum oxide are 57 nm, 103 nm, and  $60\ \mu\text{m}$ , respectively. It clearly shows homogeneous and defect-free structures.

According to the Pourbaix diagram,<sup>27</sup> theoretically,  $\text{Cu}_2\text{O}$  could be formed in a broad range of pH values between 5 and 12, while to apply optimal low current is more important, because Cu metal could be electrodeposited at high cathodic current.<sup>28</sup>

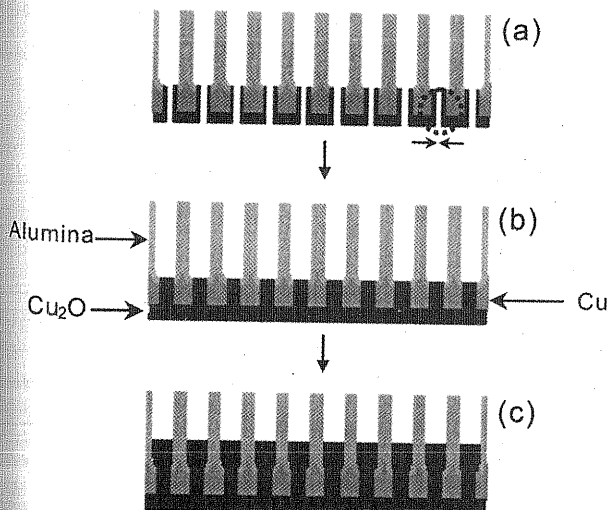


Figure 5. Schematic diagram of the production of  $\text{Cu}_2\text{O}$  nanowire by applying constant current on a porous anodic alumina membrane.

In this work, to deposit pure  $\text{Cu}_2\text{O}$  nanowires in the absence of Cu and CuO into the home-made nanoporous alumina template, an appropriate cathodic current of  $-0.5\ \text{mA}/\text{cm}^2$  was applied constantly for 4 h. For the confirmation of complete filling of  $\text{Cu}_2\text{O}$  into an alumina template with uniform orientation, SEM and TEM analysis were performed. For the TEM measurement, ion milling was applied to make an ultrathin film of  $\text{Cu}_2\text{O}$  nanowires/ $\text{Al}_2\text{O}_3$  templates via removal of a blank alumina membrane with the oxide thickness of  $55\ \mu\text{m}$  at the upper part and Cu substrate deposited at the bottom part. Figure 3 is a morphological observation of cross-sectional view (a: SEM image) and top-view (b, c: TEM image) of the  $\text{Cu}_2\text{O}$  deposited in the alumina template. In Fig. 3a, one can clearly see  $\text{Cu}_2\text{O}$  nanowires filling the alumina membrane. To confirm that  $\text{Cu}_2\text{O}$  had completely filled the alumina template with a uniform orientation, TEM analysis was performed. Compact filling of defect-free  $\text{Cu}_2\text{O}$  nanowires of the pores is clearly shown in Fig. 3b and c. The diameter of  $\text{Cu}_2\text{O}$  nanowires marked with dotted arrow (Fig. 3c) is *ca.* 57 nm. Because the length of  $\text{Cu}_2\text{O}$  nanowires is  $4\ \mu\text{m}$ , the aspect ratio and the deposition rate is *ca.* 70 and  $1\ \mu\text{m}/\text{h}$ , respectively. EDS coupled with TEM instrument analyzed atomic concentration of nanowires of internal pore and the atomic percent of Cu is twice the atomic percent of O, confirming the fabrication of  $\text{Cu}_2\text{O}$  nanowires.

Figure 4 shows the XRD analysis of the  $\text{Cu}_2\text{O}$  nanowires deposited into an ordered alumina membrane.  $\text{Cu}_2\text{O}$  was preferentially grown with a [100] fiber texture. Generally, JCPDS card file 5-667 for chemically prepared  $\text{Cu}_2\text{O}$ <sup>29</sup> presents different relative intensity patterns of  $\text{Cu}_2\text{O}$  crystalline peaks. When the relative intensity of the 111 peak is 100%, the 200 peak has only 37% relative intensity value. Interestingly, our observations are different, *i.e.*, the 200 peak has the highest intensity, with lower intensity 111, 220, and 311 peaks. We assume that this phenomenon is strongly affected by the preparation method, especially the solution pH. Recently, Golden *et al.*<sup>30</sup> demonstrated that films of  $\text{Cu}_2\text{O}$  with a strong (200) sheet texture are selectively formed in pH 9, while a higher pH value of 12 induced a (111) sheet texture. Therefore, our observations are in agreement with previous work. However, a complete understanding of the deposition behavior of single-crystalline  $\text{Cu}_2\text{O}$  is still lacking and, therefore, further investigation should be undertaken.

Figure 5 shows a schematic diagram of the electrodeposition of single-crystalline  $\text{Cu}_2\text{O}$  with [100] fiber texture on an alumina template. Cu lactate solution was placed in contact with the Cu thin-film substrate located at the bottom of the porous alumina membrane.  $\text{Cu}_2\text{O}$  was first deposited at both conducting Cu edges of the membrane hole (see dotted circle in Fig. 5a) and then  $\text{Cu}_2\text{O}$  coalesced as shown in Fig. 5b.  $\text{Cu}_2\text{O}$  grew continuously to the open top pore (Fig. 5c).

At present, we succeeded only in the production of  $\text{Cu}_2\text{O}$  nanowires with a length of  $4\ \mu\text{m}$  and this would limit its application in solar cells and sensors. Thus, we will try to prepare longer  $\text{Cu}_2\text{O}$  nanowires with higher aspect ratio and improved compactness. We expect that this paper will be a reference for the development of better nanoscale solar cell systems.

#### Acknowledgments

This work was supported by Korea Science and Engineering Foundation via Research Center for Energy Conversion and Storage and also by Inha University Research Year program.

Dr. Lee assisted in meeting the publication costs of this article.

#### References

1. S. Iijima, *Nature (London)*, **354**, 56 (1991).
2. X. F. Duan, Y. Huang, Y. Cui, J. F. Wang, and C. M. Lieber, *Nature (London)*, **409**, 66 (2001).
3. S. J. Tans, R. M. Verschueren, and C. Dekker, *Nature (London)*, **393**, 49 (1998).
4. H. Dai, E. W. Wong, Y. Z. Lu, S. Fan, and C. M. Lieber, *Nature (London)*, **375**, 769 (1995).
5. J. Choi, J. Schilling, K. Nielsch, R. Hilebrand, M. Reiche, R. B. Wehrspohn, and U. Gösele, *Mater. Res. Soc. Symp. Proc.*, **722**, L5.2 (2002):

6. J. Malicka, I. Cryczynski, J. Kusba, Y. B. Shen, and J. R. Lakowicz, *Biochem. Biophys. Res. Commun.*, **294**, 886 (2002).
7. H. Masuda, M. Ohya, H. Asoh, M. Nakao, M. Nohtomi, and T. Tamamura, *Jpn. J. Appl. Phys., Part 2*, **38**, L1403 (1999).
8. H. Asoh, K. Nishio, M. Nakao, T. Tamamura, and H. Masuda, *J. Electrochem. Soc.*, **148**, B152 (2001).
9. M. Masuda and K. Fukuda, *Science*, **266**, 1466 (1995).
10. A. P. Li, F. Müller, A. Birner, K. Nielsch, and U. Gösele, *J. Appl. Phys.*, **84**, 6023 (1998).
11. H. Masuda, K. Yada, and A. Osaka, *Jpn. J. Appl. Phys., Part 2*, **37**, L1340 (1998).
12. H. Masuda, M. Ohya, H. Asoh, and K. Nishio, *Jpn. J. Appl. Phys., Part 2*, **40**, L1217 (2001).
13. K. Nielsch, J. Choi, K. Schwirn, R. B. Wehrspohn, and U. Gösele, *Nano Lett.*, **2**, 677 (2002).
14. J. Choi, G. Sauer, K. Nielsch, R. B. Wehrspohn, and U. Gösele, *Chem. Mater.*, **15**, 776 (2003).
15. K. Nielsch, F. Müller, A. P. Li, and U. Gösele, *Adv. Mater. (Weinheim, Ger.)*, **12**, 582 (2000).
16. Y. Zhang, G. Li, Y. Wu, B. Zhang, W. Song, and L. Zhang, *Adv. Mater. (Weinheim, Ger.)*, **14**, 1227 (2002).
17. M. S. Sander, A. L. Prieto, R. Gronsky, T. Sands, and A. M. Stacy, *Adv. Mater. (Weinheim, Ger.)*, **14**, 665 (2002).
18. M. S. Sander, R. Gronsky, T. Sands, and A. M. Stacy, *Chem. Mater.*, **15**, 335 (2003).
19. J. Lee, J. Oh, and S. H. Woo, *Acta Mater.*, Submitted.
20. M. Steinhart, J. H. Wendorff, A. Greiner, R. B. Wehrspohn, K. Nielsch, J. Schilling, J. Choi, and U. Gösele, *Science*, **296**, 1997 (2002).
21. J. Lee and Y. Tak, *Electrochem. Solid-State Lett.*, **4**, C63 (2001).
22. P. E. de Jongh, D. Vanmaekelbergh, and J. J. Keely, *Chem. Mater.*, **11**, 3512 (1999).
23. P. E. de Jongh, D. Vanmaekelbergh, and J. J. Keely, *Chem. Commun. (Cambridge)*, **1999**, 1069.
24. M. Hara, T. Konto, M. Komoda, S. Ikeda, K. Shinohara, A. Tanaka, J. N. Kondo, and K. Domen, *Chem. Commun. (Cambridge)*, **1998**, 2185.
25. A. O. Musa, T. Akomolafe, and M. J. Carter, *Sol. Energy Mater. Sol. Cells*, **51**, 305 (1998).
26. K. Santra, P. Chatterjee, and S. P. Sen Gupta, *Sol. Energy Mater. Sol. Cells*, **57**, 345 (1999).
27. M. Pourbaix, *Atlas of Electrochemical Equilibria in Aqueous Solutions*, NACE, Houston, TX (1974).
28. J. Lee and Y. Tak, *Electrochem. Solid-State Lett.*, **3**, 69 (2000).
29. Joint Committee on Powder Diffraction Standards, Swarthmore, PA 19081.
30. T. D. Golden, M. G. Shumsky, Y. Zhou, R. A. VanderWerf, R. A. Van Leeuwen, and J. A. Switzer, *Chem. Mater.*, **8**, 2499 (1996).

C  
integ  
or "s  
such  
ditive  
Moni  
well &  
the cc  
El  
been  
electro  
knowi  
domai  
chlori  
(PEG)  
the ba  
Th  
ogy, a  
are no  
tion ha  
curves  
few stu  
the d  
degrad  
(EIS)<sup>9</sup>.  
experir  
the deg  
on the  
In ti  
posit m  
trode w  
ditives.  
follow  
scene p

Copi  
out fron  
The bat  
three ad  
containi  
B here)  
ditive C

\* Electr  
\* E-mai