

4. The substitution of neon for helium as the buffer gas resulted in a decrease (by nearly an order of magnitude) in the intensity of the emission spectral lines from the 9p, 8d, 6f, and 6g levels in the case of C II (or the 6p, 5d, and 4f levels in the case of Zn II), which can be populated as a result of a charge-exchange reaction involving He<sup>+</sup>.

It was shown that the time evolution of the emission changes for these lines. In the case of neon, there was no prolonged afterglow following the cutoff of the current, as is typical of experiments in which He was used as the buffer gas. In the neon case we observed only a brief recombination flash.

The experimental data show directly that the population of the metal ions is a consequence of charge exchange of metal atoms with ions of the buffer gas. Presumably other plasma-chemical reactions could also occur.

5. It was found that the optimum length of the laser pulse was  $\tau_D \sim 3 \mu\text{s}$ . The plasma expansion velocity  $v_{pl}$  was  $\sim 4 \cdot 10^5 \text{ cm/s}$  at  $I_p = 100 \text{ A}$  and buffer-gas pressure  $P \sim 3 \text{ torr}$ . Over the time the plasma expands to  $\sim 10\text{-}15 \text{ mm}$ , and after the current cutoff, the plasma begins to cooloff. Under these conditions, estimates of the cooling rate show that the cooling due to elastic collisions is at least an order of magnitude more effective than the cooling due to the expansion.

On the basis of the experimental results presented above, we can propose the following mechanism for the formation of active media in lasers with a pulsed plasma source of metal vapor.

During the application of the high-voltage pulse to the outer electrodes of the active element of the laser, a plasma consisting of the metal vapor and the buffer gas is produced in each gap between plates. The plasma expands into the preionized buffer gas. Metal ions are formed as a result of a charge exchange with He<sup>+</sup> or Ne<sup>+</sup> ions. After the current ends, the plasma cools off in collisions with the buffer gas.

The mechanism responsible for the formation of the population inversion must be examined separately for each specific chemical element and spectral transition.

In summary, this study has shown that the following physical processes are observed during the formation of the active medium of an SPER laser: a) preionization of the buffer gas; b) spreading of the current through the volume of expanding plasma; c) mixing of the metal vapor and the buffer gas as the plasma expands; d) plasma-chemical reactions in the expanding plasma; e) a predominately collisional mechanism for the cooling of the plasma.

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## InAsSb/GaAlAsSb avalanche photodiode with separate absorption and multiplication regions

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*Pis'ma Zh. Tekh. Fiz.* **14**, 986-991 (June 12, 1988)

Preliminary results of our study of the avalanche multiplication of the photocurrent in heterostructures based on multicomponent GaInAsSb/Sb solid solutions were reported in Ref. 1. It is shown that these substances hold promise for development of avalanche photodiodes for the spectral interval 1.6-2.5  $\mu\text{m}$ , which is of interest (for example) for third-generation optical-fiber communications.<sup>2</sup>

It was shown in Ref. 1 that the ratio of the electron ionization coefficients in GaInAsSb solid solutions is  $\beta/\alpha \approx 5\text{-}10$ . Multiplication factors of 20 at room temperature and 50-100 at liquid nitrogen temperature were achieved. However, the dark current in n-GaInAsSb-p-GaAlAsSb heterostructures in the prebreakdown region, which is determined by impurity and interband tunneling in the narrow-gap substance, hinders the development of efficient avalanche photodiodes for operation at room temperature.

Susa et al.<sup>3</sup> have suggested a structure with separate absorption and multiplication regions, in

a narrow-gap substance. The construction was first implemented in an InGaAs/InP avalanche photodiode for the spectral interval 1-1.6  $\mu\text{m}$  (Refs. 3 and 4).

The placement of the multiplication region with the p-n junction in a wide-gap substance makes it possible to reduce the electric field at the heterojunction with the narrow-gap region and to avoid significant tunneling currents in this region when avalanche breakdown is reached in the multiplication region. As was shown in Ref. 5, the carrier concentrations in the layers and the layer thicknesses must meet some strict requirements.

The construction proposed by Susa et al.<sup>3</sup> was used in the present study to develop an avalanche photodiode on the basis of GaInAsSb/GaAlAsSb solid solutions. A narrow-gap absorption region was made from an n-GaInAsSb solid solution ( $E_g = 0.52 \text{ eV}$ ), while the wide-gap n-p junction was in GaAlAsSb of indirect-gap composition ( $E_g = 1.2 \text{ eV}$ ), separated from the heterojunction by 0.3  $\mu\text{m}$ . The structures were fabricated by liquid-phase epitaxy on a n-type

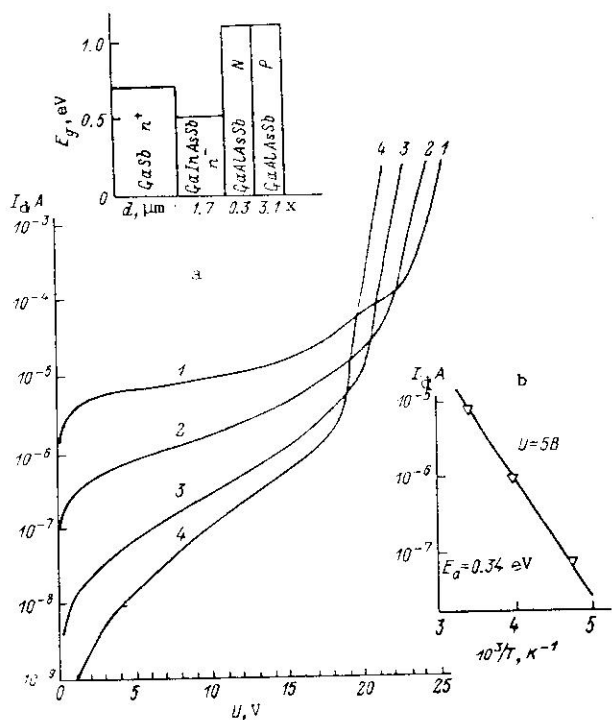


FIG. 1. Reverse branch of the current-voltage characteristic of an avalanche photodiode made from GaInAsSb/GaAlAsSb with separate absorption and multiplication regions, at various temperatures  $T$  (K): 1) 296; 2) 253; 3) 213; 4) 173. Inset a: Distribution of the width of the bandgap along the coordinate in the avalanche-photodiode structure. b: Temperature dependence of the reverse dark current at  $U_{rev} = 5$  V.

GaSb substrate. The avalanche photodiodes were fabricated as mesa diodes with a sensitive area having a diameter  $\phi \approx 280$   $\mu\text{m}$ . The carrier concentration in the narrow-gap region was  $(5-7) \cdot 10^{15}$   $\text{cm}^{-3}$ ; that in the narrow-gap region was  $7 \cdot 10^{16}$   $\text{cm}^{-3}$ ; and the  $p^+$  layer of GaAlAsSb was doped with Ge to a concentration of  $5 \cdot 10^{18}$   $\text{cm}^{-3}$ . The layer thicknesses are shown in Fig. 1.

A study of the current-voltage characteristics of the diodes over the temperature range 296-78 K showed that the temperature coefficient of the breakdown voltage was positive:  $\gamma = (\Delta U_B / U_B)(1/\Delta T) > 0$ ;  $\gamma = 2 \cdot 10^{-3}$   $\text{K}^{-1}$ . That is, the breakdown was of an avalanche nature (Fig. 1). The breakdown voltage was determined by the wide-gap substance and was 21-24 V at  $T = 296$  K and 18-20 V at 78 K. The dark current density at  $U = 0.5 U_B$  was  $j = 1-2 \cdot 10^{-2}$   $\text{A}/\text{cm}^2$ , which is lower by a factor of two or three than that in avalanche-photodiode structures of the ordinary type, which we described in Ref. 1.

The dark current in the voltage interval 5-12 V was determined by generation and recombination in the layer of narrow-gap material. The activation energy in this interval was  $E_a = 0.34$  eV, according to the temperature dependence of the reverse current (Fig. 1b). The electric field in the space-charge region of the wide-gap n-p junction was  $E = 3.5 \cdot 10^5$  V/cm, and it fell off to  $8 \cdot 10^4$  V/cm at the boundary with the narrow-gap layer. The field penetrated into the narrow-gap layer when the voltage across the structure was about 4-5 V, as is quite evident from the step on the reverse branch of the current-voltage characteristic with illumination. A significant multiplication of the photocurrent

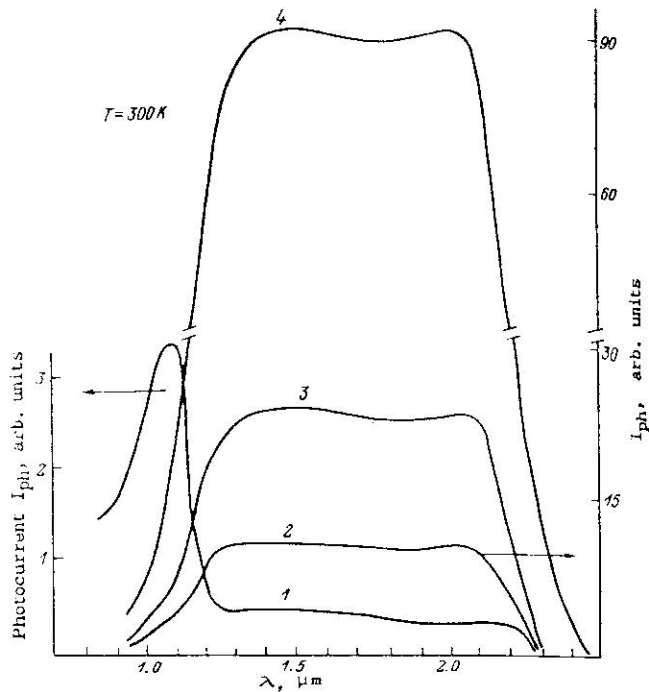


FIG. 2. Photocurrent spectrum in a GaInAsSb/GaAlAsSb avalanche photodiode at various several voltages ( $T = 296$  K).  $U$  (V): 1) 0; 2) 5; 3) 10; 4) 20.

was observed even at  $V \geq 6$  V. According to estimates, the field required for the beginning of significant tunneling in the wide-gap substance was  $E \sim 10^6$  V/cm, while that in the narrow-gap substance was above  $2.5 \cdot 10^5$  V/cm.

Figure 2 shows the photosensitivity spectrum at various voltages. Since the diode was illuminated through the wide-gap p-type layer, the photosensitivity at zero bias as determined primarily by the GaAlAsSb (curve 1). When the voltage was raised above 4 V, and the electric field penetrated into the narrow-gap layer, the photosensitivity spectrum assumed a shape characteristic of a heterostructure. The falloff at the long wavelength and short wavelength ends corresponded to the band-gap widths of the wide-gap and narrow-gap materials. A significant increase in the photosensitivity was observed corresponding to multiplication coefficients of 30-50 at room temperature. At  $T = 78$  K, the coefficients were  $M = 200-500$  in the high-voltage region.

The results found for the lowering of the tunneling component of the reverse current in the pre-breakdown region and the realization of an avalanche breakdown in our structures are related to not only the separation of the absorption and multiplication regions but also the use of the indirect-gap material GaAlAsSb in the multiplication region. This approach made it possible to reduce the tunneling probability to a level substantially lower than that in a direct-gap substance having a band gap of the same width, as a result of the participation of phonons in the process.<sup>6</sup> Under this condition, the requirement on the carrier concentration in the multiplication region may be less stringent than that in Ref. 3 and 5. Further evidence for this conclusion comes from a study of the current-voltage characteristics of GaAlAsSb diode structures of the same composition which we fabricated specifically for use as reference structures.

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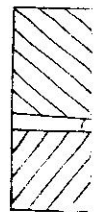
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Improving these avalanche photodiodes further require optimizing several parameters of the structure, including the carrier concentration in the wide-gap layer and the layer thicknesses, which determine the electric field at the heterojunction. It is important to reduce the field to  $E \leq 6 \cdot 10^4$  V/cm in order to avoid an additional multiplication in the absorption layer. Furthermore, "resonant" or quasi-resonant compositions of the wide-gap GaAlAsSb solid solution (with  $E_g \approx \Delta_0$ ) could be used to make the multiplication region, in order to reduce the excess-gain coefficient. The use of these compositions would permit higher values of the hole and electron ionization coefficients.<sup>7</sup>

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## Frequency conversion in single and series-connected end-type Josephson junctions

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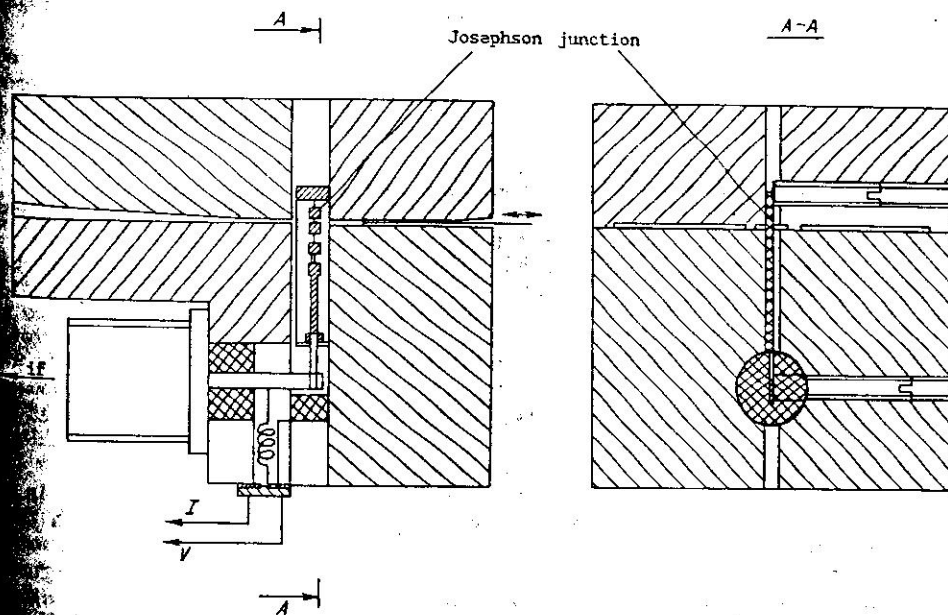
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Cooled frequency converters are used in meter-submillimeter-range superheterodyne receivers in order to raise the sensitivity.<sup>1</sup> Josephson converters with an external heterodyne are suitable in principle of amplifying a signal carried at intermediate frequency if the intrinsic single-junction noise temperature is about 20T, where T is the physical temperature of the Josephson junctions (2, 3). The use of chains of identical series-connected Josephson junctions expands the dynamic range of a receiver. In the present letter we report on the study of an external-heterodyne frequency converter for the 2.2 mm range, consisting of end-

type Josephson junctions and chains of such junctions, made of refractory materials. Josephson junctions and chains of this type have been used previously<sup>4,5</sup> in a detector receiver for the 4-8-mm range. We know of no previous work on external-heterodyne frequency converters made from end-type Josephson junctions.

2. End-type Nb/a-Si/Nb Josephson junctions with a direct conductivity and an area  $S \leq 1 \mu\text{m}^2$  and chains 0.2 mm long consisting of 11 identical junctions were fabricated on silicon wafers with a thickness  $t = 0.4$  mm by the procedure described



Construction of the frequency converter.